# Refinement of Barium Dititanate 

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#### Abstract

BaTi}_{2} \mathrm{O}_{5}\), monoclinic, $A 2 / m, a=9.409$ (3), $b=3.932$ (1), $c=16.907$ (5) $\AA, \beta=103^{\circ} 5$ (2)', $Z=6$, $D_{x}=5.12 \mathrm{~g} \mathrm{~cm}^{-3}$. Crystals were prepared by rapidly cooling a melt of composition $\mathrm{BaO}: \mathrm{TiO}_{2}=30: 70$ mol. \%. Edge-sharing titanium coordination octahedra form infinite zigzag chains parallel to $\mathbf{b}$. The structure can also be described as an almost closepacked array of barium and oxygen atoms with titanium atoms in octahedral voids. The direction of packing is approximately [ $\overline{1} 3 \overline{1}$ ] and the stacking sequence is $A B C A B C A B C . \mathrm{TiO}_{6}$ octahedra show appreciable distortions similar to those observed in $\mathrm{PbTiO}_{3}$.


Introduction. A crystal with dimensions approximately $0.07 \times 0.04 \times 0.2 \mathrm{~mm}$ was used to measure the cell parameters and intensities. The cell constants were refined by a least-squares fit to the angular settings of 12 reflexions with $25^{\circ} \leq \theta \leq 30^{\circ}$ (Mo $K \alpha_{1}, \lambda=0.70926 \AA$ ). The systematic absences are $h k l$ for $k+l$ odd. The intensities of 6691 reflexions with $\theta \leq 38^{\circ}$ were collected on an automatic four-circle diffractometer with Zr filtered Mo $K \alpha$ radiation, $\theta-2 \theta$ step-scan mode with 60 steps of $0.01^{\circ}$ and 2 s counting time per step. The background was measured for 10 s at each end of the scanning interval. After correction for absorption [ $\mu(\mathrm{Mo} K \alpha)=136.9 \mathrm{~cm}^{-1}$, transmission factors between 0.33 and 0.62 ] reduction and averaging of the intensities resulted in 1843 unique structure factors, 431 of which were considered to be of zero intensity ( $I<2 \sigma_{I}$ ). The structure was refined by full-matrix least-squares calculations starting from the atomic coordinates given
by Harrison (1956). Scattering factors were taken from International Tables for X-ray Crystallography (1962); the observed structure factors were weighted according to $w=1 / \sigma^{2}(F)$. Other experimental procedures and the computer programs used are the same as given by Tillmanns \& Gebert (1973).

The final weighted residual $R_{2}=\left(\sum w(\Delta F)^{2} / \sum w \cdot F_{o}^{2}\right)^{1 / 2}$ was 0.031 , the corresponding unweighted $R_{1}$ was 0.026 for observed reflexions only and 0.072 including the 431 unobserved reflexions with $F_{o}=0$. The resulting positional and thermal parameters are given in Table 1.*

Discussion. The structural principle of barium dititanate has already been determined by Harrison (1956) from two-dimensional photographic data. The proposed structure was essentially correct although the spread in Ti-O distances was large and there was some uncertainty about the position of $O(4)$ because of diffraction ripples around the Ba atoms. Fig. 1 shows a projection of the structure along $\mathbf{b}$. Edge-sharing $\mathrm{TiO}_{6}$ octahedra form planar groups of three octahedra each, which are linked by common corners to other groups in the plane (010). In the direction of $\mathbf{b}$, infinite zigzag chains are formed via common octahedral edges.

The two independent Ba atoms in the asymmetric unit have a cuboctahedral 12-coordination with mean $\mathrm{Ba}-\mathrm{O}$ distances of 2.929 and $2.884 \AA$ respectively ( $c f$.

[^0]Table 1. Fractional atomic coordinates and vibrational parameters $\left(\AA^{2}\right)$ with the significant figures of the estimated standard deviations in parentheses
All atoms are situated on the mirror plane at $y=0$. The definition of the Debye-Waller temperature factor is $\exp \left[-\frac{1}{4}\left(h^{2} a^{* 2} B_{11}\right.\right.$ $\left.\left.+k^{2} b^{* 2} B_{22}+l^{2} c^{* 2} B_{33}+2 h l a^{*} c^{*} B_{13}\right)\right]$.

|  | $x / a$ | $z / c$ | $B_{11}$ | $B_{22}$ | $B_{33}$ | $B_{13}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ba}(1)$ | $0 \cdot 48223$ (3) | $0 \cdot 13107$ (2) | $0 \cdot 70$ (1) | $0 \cdot 49$ (1) | $0 \cdot 62$ (1) | $0 \cdot 18$ (1) |
| $\mathrm{Ba}(2)$ | 0.0 | 0.0 | $0 \cdot 54$ (1) | $0 \cdot 43$ (1) | $0 \cdot 67$ (1) | $0 \cdot 09$ (1) |
| $\mathrm{Ti}(1)$ | $0 \cdot 19441$ (10) | 0.66610 (6) | $0 \cdot 32$ (3) | $0 \cdot 17$ (3) | $0 \cdot 14$ (3) | $0 \cdot 13$ (3) |
| Ti(2) | $0 \cdot 12731$ (10) | $0 \cdot 29292$ (6) | $0 \cdot 22$ (3) | $0 \cdot 20$ (3) | $0 \cdot 11$ (3) | -0.01 (3) |
| Ti(3) | $0 \cdot 28946$ (11) | $0 \cdot 46098$ (7) | $0 \cdot 34$ (3) | 2.07 (6) | $0 \cdot 10$ (3) | $0 \cdot 11$ (2) |
| $\mathrm{O}(1)$ | $0 \cdot 1625$ (4) | $0 \cdot 1749$ (3) | $0 \cdot 8$ (2) | 0.8 (2) | $0 \cdot 5$ (1) | $0 \cdot 3$ (2) |
| O(2) | $0 \cdot 1015$ (4) | 0.7651 (3) | $0 \cdot 1$ (1) | $0 \cdot 6$ (1) | $0 \cdot 2$ (1) | $0 \cdot 0$ (1) |
| $\mathrm{O}(3)$ | $0 \cdot 2902$ (4) | 0.9646 (3) | $0 \cdot 8$ (1) | $0 \cdot 4$ (1) | 0.7 (1) | $0 \cdot 2$ (1) |
| $\mathrm{O}(4)$ | $0 \cdot 3154$ (4) | $0 \cdot 3472$ (3) | $0 \cdot 5$ (1) | 0.4 (1) | $0 \cdot 3$ (1) | $0 \cdot 2$ (1) |
| $\mathrm{O}(5)$ | $0 \cdot 5$ | $0 \cdot 5$ | 0.0 (2) | $0 \cdot 9$ (2) | 0.6 (2) | -0.2 (2) |
| O(6) | $0 \cdot 2107$ (4) | $0 \cdot 5576$ (2) | $0 \cdot 8$ (1) | $0 \cdot 9$ (2) | $0 \cdot 3$ (1) | $0 \cdot 3$ (1) |
| O(7) | $0 \cdot 0710$ (4) | 0.3911 (3) | $0 \cdot 5$ (1) | 0.7 (2) | $0 \cdot 2$ (1) | $0 \cdot 1$ (1) |
| $\mathrm{O}(8)$ | $0 \cdot 3780$ (4) | 0.7113 (3) | $0 \cdot 5$ (1) | $0 \cdot 7$ (2) | 0.6 (1) | 0.0 (1) |

Table 2). All O atoms are also cuboctahedrally surrounded by 12 other O or Ba atoms and like most other barium titanates the structure of $\mathrm{BaTi}_{2} \mathrm{O}_{5}$ is based on a nearly close-packed array of Ba and O atoms with 9 layers stacked along [ $\overline{1} 3 \overline{1}]$ in the sequence $A B C A B C A B C$.

One interesting feature of the crystal structure is that $\mathrm{O}(8)$ is coordinated by only one Ti and three Ba atoms, thus receiving a bond strength sum (Pauling, 1960) of only $1 \cdot 17$ valence units. This extreme undersaturation of $\mathrm{O}(8)$ seems to be compensated by a pronounced distortion of the coordination octahedron around $\mathrm{Ti}(1) . \mathrm{Ti}(1)$ is displaced from the centre of the octahedron towards $\mathrm{O}(8)$; the resulting $\mathrm{Ti}(1)-\mathrm{O}(8)$ distance is $1.724 \AA$. Four Ti-O distances are in the range $1.875-2.056 \AA$, the sixth $[\mathrm{Ti}(1)-\mathrm{O}(7)]$ is considerably longer, $2 \cdot 466 \AA$. While the short $\mathrm{Ti}(1)-\mathrm{O}(8)$ distance could be interpreted with the extended electrostatic valence rule (Pauling, 1960; Baur, 1970), there is only poor agreement between other observed Ti-O distances and the distances calculated according to Baur's (1970) rule 3. O(7) for example is coordinated by three Ti and two Ba atoms and receives a bondstrength sum $p(\mathrm{O})=2.33 \mathrm{v} . \mathrm{u}$., which would correspond to calculated Ti-O distances of $2.01,1.96$ and $2.05 \AA$


Fig. 1. Projection of the structure along [010]. All atoms are located in the mirror planes at $y=0.5$ (solid lines) and $y=0$ (broken lines).
for $\mathrm{O}(7)-\mathrm{Ti}(1), \mathrm{O}(7)-\mathrm{Ti}(2)$ and $\mathrm{O}(7)-\mathrm{Ti}(3)$ respectively (Baur, 1970, rule 3). The observed distances, however, are $2 \cdot 466,1 \cdot 853$ and $2 \cdot 126 \AA$ respectively. The bondstrength sums calculated from the interatomic distances with Brown \& Shannon's (1973) empirical bond-strength-bond-length curves deviate appreciably from the atomic valences; both Ba atoms for example are overbonded by $25 \%$.

A computer simulation of the crystal structure with the distance least-squares method (Meier \& Villiger, 1969) where the prescribed interatomic distances were calculated with Baur's (1970) rule 3 and their weights were based on the electrostatic strengths of these bonds (Baur, 1972), showed that such an electrostatic model

Table 2. Interatomic distances and angles
Standard deviations are approximately $0.004 \AA$ for $\mathrm{Ba}-\mathrm{O}$ and $\mathrm{Ti}-\mathrm{O}$ distances, 0.002 for $\mathrm{Ti}-\mathrm{Ti}$ and 0.006 for $\mathrm{O}-\mathrm{O}$ distances. Bond angles have a standard deviation of $0 \cdot 2^{\circ}$.

| Central Atom: | $\mathrm{Ba}(1)$ |  | $O(1)--0(7)$ | $35.6^{\circ}$ | $2.889 \%$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ba}(1)-\mathrm{O}(8)$ |  | 2.692 月 | Central Atom: | T1(3) |  |
| $\bigcirc(8)$ |  | 2.695 (2x) |  |  |  |
| $\bigcirc(4)$ |  | 2.704 (2x) | Ti ( 3) -o (6) |  | 1.940 \% |
| -O(3) |  | 2.962 | -O(5) |  | 1.941 |
| -O(3) |  | 2.980 | -0(3) |  | 1.967 (2x) |
| -0(5) |  | 2.995 (2x) | -O(4) |  | 1.993 |
| -0(6) |  | 3.240 (2x) | -0(7) |  | 2.126 |
| -O(1) |  | 3.257 | $0(6)-0(5)$ | $105.6{ }^{\circ}$ | 3.091 |
|  |  |  | $0(6)--0(3)$ | 88.4 | 2.725 (2x) |
| Central Atom: | $\mathrm{Ba}(2)$ |  | $0(6)-0(4)$ | 165.0 | 3.900 |
|  |  |  | $0(6)-0(7)$ | 37.9 | 2.824 |
| $\mathrm{Ba}(2)-\mathrm{O}(6)$ |  | 2.805 (4x) | $0(5)--0(3)$ | 89.6 | 2.754 (2x) |
| -0(7) |  | 2.874 (4x) | $0(5)--0(4)$ | 89.4 | 2.768 |
| -0(3) |  | 2.924 (2x) | $0(5)--0(7)$ | 166.6 | 4.040 |
| -O(1) |  | 3.006 (2x) | $0(3)-0(3)$ | 176.5 | 3.932 |
|  |  |  | $0(3)-0(4)$ | 91.7 | 2.842 (2x) |
| Central Atom: | Ti(1) |  | $0(3)-0(7)$ | 90.8 | 2.916 (2x) |
|  |  |  | $0(4)-0(7)$ | 77.2 | 2.571 |
| Ti(1) $-0(3)$ |  | 1.724 |  |  |  |
| -o(6) |  | 1.375 | Central Atom: | O(1) |  |
| -O(1) |  | 1.999 (2x) |  |  |  |
| -O(2) |  | 2.056 | $\mathrm{Ti}(1)-\mathrm{Ti}(1)$ | 159.0 | 3.932 |
| -0(7) |  | 2.466 | $\mathrm{Ti}(1)-\mathrm{Ti}(2)$ | 97.4 | 3.078 (2x) |
| $0(8)-0(6)$ | $37.9^{\circ}$ | 2.716 |  |  |  |
| $0(8)-0$ ( 1 ) | 36.8 | 2.731 (2x) | Central Atom: | O(2) |  |
| $0(8)-0(2)$ | 102.0 | 2.944 |  |  |  |
| 0 (3) $-0(7)$ | 176.9 | 4.138 | Ti(2)-Ti(2) | 152.7 | 3.932 |
| $O(6)--0(1)$ | 36.3 | 2.900 (2x) | $\mathrm{Ti}(2)-\mathrm{Ti}(1)$ | 93.0 | 3.078 (2x) |
| $0(6)-0(2)$ | 160.1 | 3.372 | $\mathrm{Ti}(2)-\mathrm{Ti}(2)$ | 93.3 | 3.134 (2x) |
| $0(6)-0(7)$ | 85.2 | 2.970 | $\mathrm{Ti}(1)-\mathrm{Ti}(2)$ | 101.2 | 3.255 |
| $0(1)-0(1)$ | 159.0 | 3.932 |  |  |  |
| $0(1)--0(2)$ | 80.3 | 2.630 (2x) | Central Atom: | O(3) |  |
| $0(1)-0(7)$ | 82.7 | 2.971 (2x) |  |  |  |
| $0(2)-0(7)$ | 74.9 | 2.769 | $\mathrm{Ti}(3)-\mathrm{Ti}(3)$ | 176.5 | 3.932 |
| Central Atom: | Ti(2) |  | Central Atom: | O(4) |  |
| $\mathrm{Ti}(2)-\mathrm{O}(4)$ |  | 1.799 | $\mathrm{Ti}(2)-\mathrm{Ti}(3)$ | 99.9 | 2.905 |
| -0(7) |  | 1.853 |  |  |  |
| -0(2) |  | 2.023 (2x) | Central Atom: | O(5) |  |
| -0(1) |  | 2.096 |  |  |  |
| -0(2) |  | 2.154 | $\mathrm{Ti}(3)-\mathrm{Ti}(3)$ | 180.0 | 3.882 |
| $0(4)-0(7)$ | 39.5 | 2.571 |  |  |  |
| $0(4)-0(2)$ | 79.8 | 2.929 (2x) | Central Atom: | 0(6) |  |
| $O(4)--0(1)$ | 97.8 | 2.942 |  |  |  |
| $0(4)-0(2)$ | 176.5 | 3.952 | $\mathrm{Ti}(1)-\mathrm{Ti}(3)$ | 162.7 | 3.712 |
| $0(7)-2(2)$ | 99.5 | 2.960 (2x) |  |  |  |
| $0(7)-0(1)$ | 172.7 | 3.941 | Central Atom: | O(7) |  |
| $0(7)-0(2)$ | 87.1 | 2.769 |  |  |  |
| $0(2)-0(2)$ | 152.7 | 3.932 | $\mathrm{Ti}(2)-\mathrm{Ti}(3)$ | 93.5 | 2.905 |
| $O(2)--0(1)$ | 79.3 | 2.630 (2x) | $\mathrm{Ti}(2)-\mathrm{Ti}(1)$ | 96.8 | 3.255 |
| $0(2)-0$ (2) | 80.7 | 2.708 (2x) | $\mathrm{Ti}(3)-\mathrm{Ti}(1)$ | 169.7 | 4.574 |

could easily be constructed without a change of the unit-cell parameters.
It therefore seems that the distortions of the $\mathrm{TiO}_{6}$ octahedra in $\mathrm{BaTi}_{2} \mathrm{O}_{5}$ are not caused by the bondstrength distribution and cannot be interpreted on electrostatic grounds only. Baur (1961) pointed out that the Ti coordination in $\mathrm{PbTiO}_{3}$ (Shirane, Pepinsky \& Frazer, 1956) with Ti-O distances of $1.78,4 \times 1.98$ and $2 \cdot 38 \AA$ should be described as a $5+1$ coordination and a similar distortion of a $\mathrm{TiO}_{6}$ octahedron in $\mathrm{Na}_{2} \mathrm{Ti}_{3} \mathrm{O}_{7}$ has been interpreted by Andersson \& Wadsley (1961) as a tendency of Ti towards 5 coordination. Table 3 gives distances between Ti atoms and their six nearest O neighbours for a number of compounds

Table 3. Distances between titanium atoms and their six nearest oxygen neighbours for a number of compounds with different Ti coordinations

| Compound | Bond lengths ( $\AA$ ) |
| :---: | :---: |
| $\mathrm{Y}_{2} \mathrm{TiO}_{5}{ }^{\text {a }}$ | $\text { Ti-O: } \underset{3.89}{1 \cdot 78}, 1 \cdot 87,1 \cdot 91,2 \times 1 \cdot 94$ |
| Fresnoite | Ti-O: $1 \cdot 63,4 \times 2 \cdot 00,3.58$ |
| $\mathrm{Ba}_{2} \mathrm{TiOSi}_{2} \mathrm{O}_{7}{ }^{\text {b }}$ |  |
| Innelite | $\begin{array}{cccc} \mathrm{Ti}(1)-\mathrm{O}: & 1.62, & 1.99, \\ 2.06,3.67 \end{array}$ |
| $\mathrm{Na}_{2} \mathrm{Ba}_{3}(\mathrm{Ca}, \mathrm{Na})(\mathrm{Ba}, \mathrm{K}, \mathrm{Mn})$ | $\begin{array}{r} \mathrm{Ti}(2)-\mathrm{O}: \begin{array}{c} 1 \cdot 90,1 \cdot 92,2 \cdot 00,2 \cdot 04, \\ 2 \cdot 09,2 \cdot 16 \end{array} \end{array}$ |
| $\mathrm{Ti}_{3} \mathrm{O}_{4}\left(\mathrm{SO}_{4}\right)_{2}\left(\mathrm{Si}_{2} \mathrm{O}_{7}\right)_{2}{ }^{\text {c }}$ | $\begin{array}{r} \mathrm{Ti}(3)-\mathrm{O}: 1 \cdot 71,1 \cdot 93,1 \cdot 96,1 \cdot 98 \\ 2 \cdot 01,3 \cdot 17 \end{array}$ |
| $\mathrm{BaTi} \mathrm{F}_{2}{ }_{5}{ }^{\text {d }}$ | $\mathrm{Ti}(1)-\mathrm{O}: \begin{gathered} 1.71, \\ 2.06,2.47 \end{gathered}$ |
| $\mathrm{PbTiO}_{3}{ }^{\text {e }}$ | Ti-O: $1 \cdot 78,4 \times 1 \cdot 98,2.38$ |
| $\mathrm{Na}_{2} \mathrm{Ti}_{3} \mathrm{O}_{7}{ }^{\text {f }}$ | $\mathrm{Ti}(2)-\mathrm{O}: \underset{2 \times 1.94,2.34}{ }=$ |

References: (a) Mumme \& Wadsley (1968); (b) Moore \& Louisnathan (1969); (c) Chernov et al. (1971); (d) This work; (e) Shirane et al. (1956); ( $f$ ) Andersson \& Wadsley (1961).
with Ti in different coordinations. It can be seen that there is no sharp limit between 5 and 6 -coordinated Ti. While Ti is clearly 5 -coordinated in $\mathrm{Y}_{2} \mathrm{TiO}_{5}$, fresnoite, and partly in innelite, other Ti atoms in innelite, in $\mathrm{BaTi}_{2} \mathrm{O}_{5}, \mathrm{PbTiO}_{3}$ and $\mathrm{Na}_{2} \mathrm{Ti}_{3} \mathrm{O}_{7}$ have a $5+1$ coordination.

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# 1,6:8,13-Cyclopropanediylidene[14]annulene 

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#### Abstract

C}_{17} \mathrm{H}_{12}\), monoclinic, $C 2 / c, Z=8, a=19 \cdot 485$ (3), $b=6.812$ (2), $c=17.881$ (5) $\AA, \beta=108.52$ (2) ${ }^{\circ}, D_{m}=$ $1 \cdot 273, D_{x}=1 \cdot 277 \mathrm{~g} \mathrm{~cm}^{-3}$, m.p. $169-170^{\circ} \mathrm{C}$. Mo $K \alpha$ diffractometer data. Final $R=0.054$ on all observed amplitudes. The strain imposed by the bridges of the annulene perimeter and by the cyclopropane ring seem to have a balancing rather than a synergic effect; as a result the annulene ring is nearly planar with bond lengths close to the aromatic value.

Introduction. X-ray diffraction data were measured with Mo $K \alpha$ radiation on a Syntex $P \bar{\top}$ four-circle diffractom-


eter equipped with graphite monochromator. Cell dimensions were obtained from 26 reflexions (plus their equivalents) for $\lambda=0.71069 \AA$. Systematic absences were $h k l$ for $h+k$ odd, $h 0 l$ for $l$ odd. A crystal of dimensions $0.30 \times 0.35 \times 0.40 \mathrm{~mm}$ was used. Intensities were collected to a maximum $2 \theta$ value of $55^{\circ}$ ( $\theta-2 \theta$ scan mode, scan width $2 \cdot 1^{\circ}+\alpha_{1} \alpha_{2}$ separation, variable scan speed between 1 and $12^{\circ} \mathrm{min}^{-1}$, dead time for coincidence correction $2.1 \times 10^{-6} \mathrm{~s}$ ). The background was counted for half the total scanning time on each side of the reflexion. Three check reflexions were monitored periodically to test crystal


[^0]:    * A list of structure factors has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 30616 ( 9 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 13 White Friars, Chester CH11NZ, England.

